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Implementation of a Finite Strain Plasticity Model for Nylon 6/6 into DYNA3D

George A. Gazonas

ARL-RP-1

September 2000

A reprint from *Structures Under Shock and Impact VI*,
N. Jones and C. A. Brebbia, editors, WIT Press (2000), pp 523-535.

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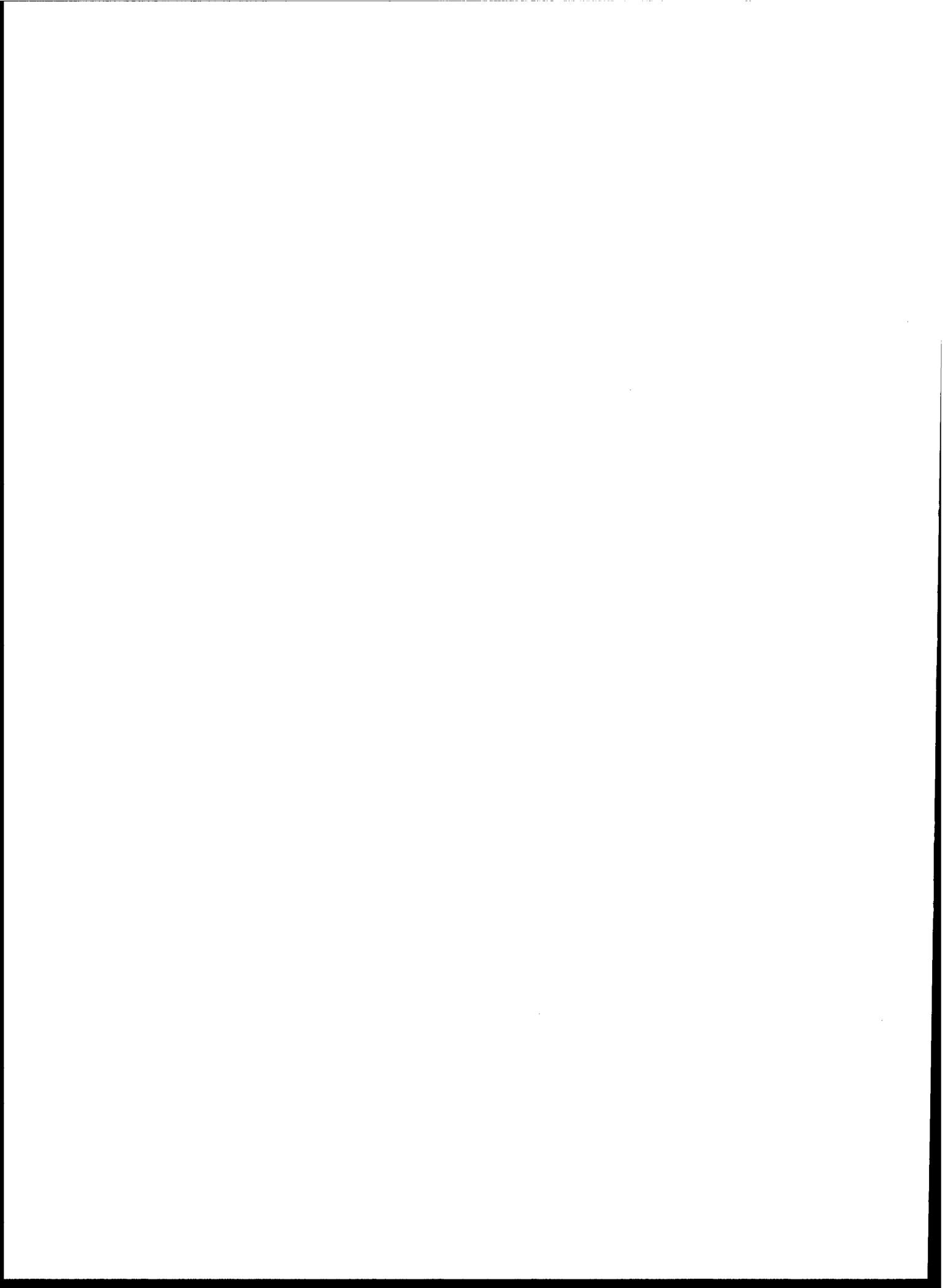
Abstract

A fully three-dimensional rate and temperature dependent finite strain plasticity model for semicrystalline nylon 6/6, originally developed for finite inelastic deformation in metals, has been implemented in the explicit, Lagrangian finite element code DYNA3D. The model has been previously implemented into the implicit code NIKE2D for comparing the engagement process of nylon obturators relative to that of copper obturators. The DYNA3D material driver was used to verify model implementation for constant strain rate input histories by comparing true stress versus true strain results with predictions from Bammann's closed-form model and NIKE2D implicit finite element results. The material model is being used for modeling the interaction and erosion of the nylon 6/6 obturator band against a smoothbore gun tube during launch in 120-mm tank rounds. The adaptive slideline feature also permits modeling complex impact and penetration phenomena by calibrating the Lagrangian element erosion strains against carefully controlled penetration experiments into nylon 6/6 plates.

Relevance to the Army

Many kinetic energy projectiles rely on nylon-based materials to form an effective high-pressure seal against hot, rapidly expanding propellant gases. The seal is effected by obturator bands, which undergo high-rate pressurization, finite strain deformation, abrasion, and wear as they travel along gun tubes. A software model of the nylon material has been developed that is being used to predict the interaction and erosion of the nylon obturator band against a smoothbore gun tube during launch in 120-mm tank rounds. This software model will give designers information on tube wear that will enable them to minimize the wear on projectile systems.

This paper provides an overview of finite strain plasticity theory, describes the implementation of the model in the Lagrangian finite-element code DYNA3D, discusses model validation, and illustrates some simulation results.



Implementation of a finite strain plasticity model for Nylon 6/6 into DYNA3D

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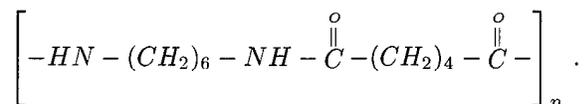
Abstract

A fully three-dimensional rate and temperature dependent finite strain plasticity model for semicrystalline nylon 6/6, originally developed for finite inelastic deformation in metals [2],[4], has been implemented in the explicit, Lagrangian finite element code DYNA3D [16]. The model has been previously implemented into the implicit code NIKE2D [10] for comparing the engagement process of nylon obturators relative to that of copper obturators [14]. The DYNA3D material driver was used to verify model implementation for constant strain rate input histories by comparing true stress versus true strain results with predictions from Bammann's closed-form model and NIKE2D implicit finite element results. The material model is being used for modeling the interaction and erosion of the nylon 6/6 obturator band against a smoothbore gun tube during launch in 120-mm tank rounds. The adaptive slideline feature also permits modeling complex impact and penetration phenomena by calibrating the Lagrangian element erosion strains against carefully controlled penetration experiments into nylon 6/6 plates.

1 Introduction

Modern kinetic energy projectiles rely upon nylon-based materials to form an effective high-pressure seal against hot, rapidly expanding propellant gases. The nylon obturator band undergoes high-rate pressurization, finite strain deformation, abrasion and wear as it travels along the gun-tube during the transient launch history lasting on the order of 10 *ms*. The oldest

and most versatile form of nylon, Dupont Zytel[®] 101 or nylon 6/6, is the most commonly used nylon obturator material and is made by polymerization of hexamethylenediamine and adipic acid [7],[8], each of which contains 6 carbon atoms. Nylon 6/6 is a linear polymer consisting of monomers joined end-to-end in a line or chain, n-times,



The binding energy between the chains are due to van Der Waals forces which are relatively weak, and the material easily softens as temperature is increased. Semicrystalline nylon 6/6 has excellent obturation properties by combining strength, moderate stiffness, high toughness, high resistance to impact and a relatively low coefficient of friction and abrasion resistance. In addition, nylon 6/6 is easily injection molded into a variety of shapes and sizes. The material is hygroscopic, readily absorbs moisture from the air, and equilibrates at 2.5% water at 50% relative humidity, and about 8.5% at 100% relative humidity [8]. This property can be problematic for use of nylon 6/6 in structures requiring tight tolerances such as tank rounds. This paper provides an overview of the finite strain plasticity theory, implementation of the model into DYNA3D, model validation, and illustration of some simulation results.

2 Constitutive assumptions

The finite strain plasticity model developed by Bammann [2] for characterizing the inelastic deformation of metals has been previously implemented in DYNA3D (model 38) using a set of internal state variables whose evolution is based upon micromechanical considerations. Both rate and temperature dependence is included in the constitutive description with plastic work being converted to heat which provides a mechanism for local material softening. Even though the microphenomenological deformation mechanisms in semicrystalline nylon differ considerably from that of metals, the gross deformation behavior of nylon appears "metal-like" in that nylon possesses strong rate and temperature dependence, and pressure independent yield behavior [15]. The original viscoplastic framework proposed by Bammann [2] is based upon the microphysical deformation mechanisms observed in metals, but the current nylon model is purely macrophenomenological in nature, and underscores the robustness of the general model. A full description of the large-deformation kinematics and the microscale to macroscale transition, based on dislocation dynamics, can be found in Bammann and Aifantis [3] where the deformation gradient is written as a multiplicative decomposition of elastic and plastic parts. The following kinematic/constitutive exposition essentially follows that found in [4]. The assumption of linear elasticity associated with this decomposition is given by,

$$\overset{\circ}{\sigma} = \lambda \text{tr}(D^e)1 + 2\mu D^e, \quad (1)$$

where λ , and μ are the Lamé parameters, and the Cauchy stress σ is connected with the elastic spin W^e ,

$$\overset{\circ}{\sigma} = \sigma - W^e \sigma + \sigma W^e. \quad (2)$$

Decomposing the velocity gradient $L = D + W$ into elastic and plastic parts, gives the elastic stretch D^e ,

$$D^e = D - D^p, \quad (3)$$

and the elastic spin W^e ,

$$W^e = W - W^p. \quad (4)$$

Within this framework, an additional equation is needed to describe the plastic spin W^p , which for the Jaumann stress rate (almost exclusively used in DYNA3D), implies that $W^p = 0$ (e.g., see discussion in reference [5]). Unfortunately, although the Jaumann stress rate is objective, it predicts a nonphysical oscillatory shear stress for bodies undergoing simple shear [9],[12] (Figure 1). In contrast to this behavior, the choice,

$$W^p = \frac{1}{2} R \left[\dot{U} U^{-1} - U^{-1} \dot{U} \right] R^T \quad (5)$$

recovers the objective Green-Naghdi stress rate [5] and predicts a monotonic increase in stress in bodies undergoing simple shear (Figure 2), with a minor correction to the equation in the original references [4],[5]. R and U are the rotation and right stretch tensors, respectively. The amplitude of dimensionless shear stress is nearly the same for both stress rate definitions to nearly 75% shear strain, compare e.g. Figures (1) and (2), which validates the implementation into DYNA3D.

The deviatoric flow rule is assumed to be of the form,

$$D^p = f(\theta) \sinh \left[\frac{|\xi| - \kappa - Y(\theta)}{V(\theta)} \right] \text{sgn}(\xi'), \quad \xi' = \sigma' - \alpha' \quad (6)$$

where θ is the temperature, k is a scalar hardening parameter, and ξ' is the difference between deviatoric Cauchy stress σ' and the tensor hardening variable α' . The evolution of the internal state variables α' and k are couched in a hardening minus recovery format,

$$\overset{\circ}{\alpha} = h(\theta) D^p - [r_d(\theta) |D^p| + r_s(\theta) |\alpha|] \alpha, \quad (7)$$

and,

$$\dot{k} = H(\theta) |D^p| - [R_d(\theta) |D^p| + R_s(\theta)] \kappa^2. \quad (8)$$

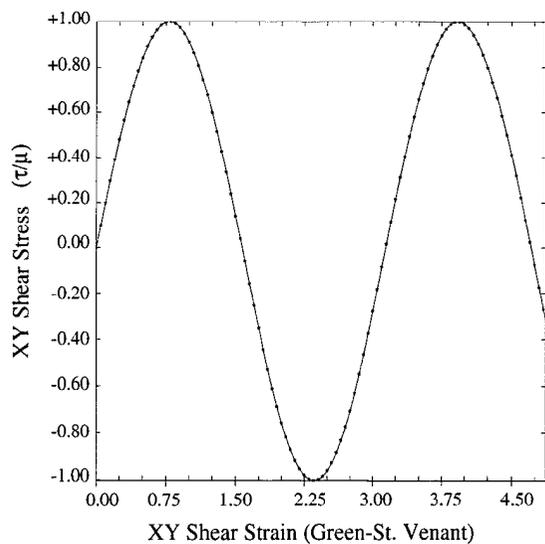


Figure 1: Dimensionless simple shear of a hypoelastic material (Jaumann stress rate); comparison of DYN3D result (***) with closed-form result (—) of Dienes [9].

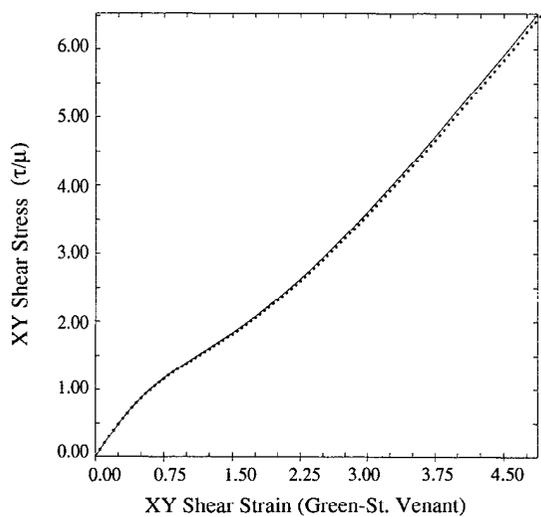


Figure 2: Dimensionless simple shear of a hypoelastic material (Green-Naghdi stress rate); comparison of new DYN3D implementation (***) with closed-form result (—) of Dienes [9].

The general theory admits nine temperature-dependent scalar exponential functions for yield $f(\theta)$, $Y(\theta)$, $V(\theta)$; hardening $h(\theta)$ and $H(\theta)$, and static/dynamic recovery $r_d(\theta)$, $r_s(\theta)$, $R_d(\theta)$, and $R_s(\theta)$. These equations are not reproduced here, but can be found in [5]. Heat flow is often neglected in simulating the physics of media undergoing rapid deformation, yet plastic work induces a local temperature increase that softens the material using the empirical expression,

$$\dot{\theta} = \frac{.9}{\rho C_v} (\sigma \cdot D^p), \quad (9)$$

where ρ is the material density, and C_v is the specific heat.

2.1 Numerical implementation

For implementation into explicit codes such as DYNA3D (1994 version), deformation rates are used to predict stress, hence Eqn. (6) is inverted for stress. In addition, the theory is considerably simplified for application to nylon 6/6. In particular, the internal state variable evolution equations are simplified by removing the temperature dependence of the hardening functions $h(\theta) = k_1$, $H(\theta) = k_2$, by setting them equal to constant values. If all the recovery functions in (7) and (8) are set to zero, the theory predicts rate-dependent bilinear hardening behavior [4]. For application to nylon 6/6, two of the static/dynamic recovery functions $r_s(\theta) = R_d(\theta) = 0$ are set to zero. Finally, the deformation is restricted to purely kinematic hardening yield behavior. These modifications are necessary for empirically fitting experimental data obtained by Kawahara [13] to the theory outlined above. The details involving fitting the various parameters to the model are lengthy, and the interested reader is referred to the discussion in [4] for more specifics.

The numerical implementation into DYNA3D is verified using the code's material driver capability that permits evaluation of the constitutive behavior under specific user-defined load paths. For a bar in uniaxial tension under constant strain rate loading, excellent agreement is found between Bammann's closed-form result, an implicit implementation into NIKE2D, and the current explicit implementation in DYNA3D (Figure 3) at two temperatures 68 °F and 392 °F and 2% humidity. The strong temperature dependence of the flow stress seen in Figure (3) is governed principally by the temperature-dependent functions in Equation (6), rather than through those in the internal state variable evolution Equations (7) and (8).

3 Simulation results

The nylon 6/6 model is currently being used to simulate transient launch dynamics in 120 mm tank rounds. Rapid pressurization and deformation of the aft portion of a kinetic energy projectile (Figure 4) accelerates the projectile downbore, causing localized deformation, abrasion, and wear of

**DYNA3D Tension of Nylon 6/6 at 100 1/s,
293 K & 473 K, and 2 % humidity.**

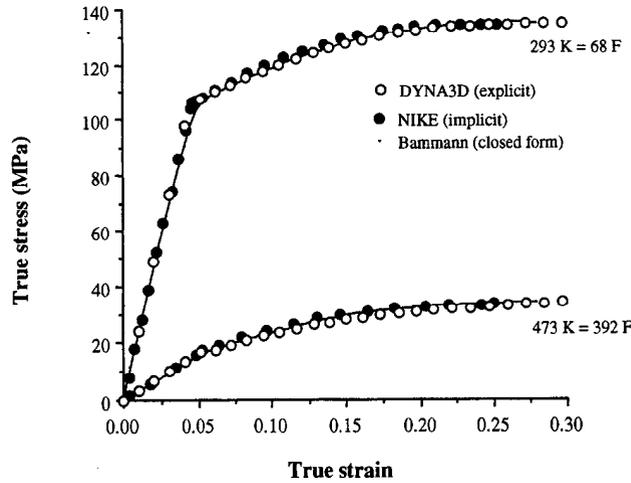


Figure 3: Validation of nylon 6/6 model implementation.

the nylon 6/6 obturator and other portions of the projectile in contact with the gun tube. Inclusion of the nylon 6/6 material model smoothes velocity transients seen within the first 4 *ms* of launch relative to prior models with linear elastic obturators (Figure 5). Despite this modeling improvement, a quantitative model for characterizing the erosion of the Lagrangian elements subject to abrasion in the gun tube is lacking. This information is needed for properly characterizing the frictional losses experienced by the projectile during the launch history.

An effort to improve our understanding of nylon 6/6 wear and erosion is currently underway using carefully controlled impact/penetration experiments into nylon 6/6 plates of various thickness and degree of humidity. An initial finite element model of an 8.26 *g* elastic steel ball, 0.5" in diameter, $V_0 = 3$ km/s impacting a nylon 6/6 plate, 1 *in* thick, 12 x 12 *in* square, and weighing 2.68 *kg* consists of 87,374 nodes and 76,992 hexahedral elements, and costs about 1 CPU hr/10 solution μs on a Cray C90 (Figure 6). Crater topology, ballistic limit, exit velocity, exit crater topology and other impact/penetration metrics are strongly dependent upon the user-defined, equivalent plastic erosion strain $\bar{\epsilon}^p$ level chosen in a particular simulation (Figure 7), (Figure 8), (Figure 9), hence it is imperative that carefully controlled impact/penetration tests be conducted to calibrate erosion strain. The slidesurfaces with adaptive new definitions (SAND) feature in DYNA3D was also modified to permit the penetration calculations.

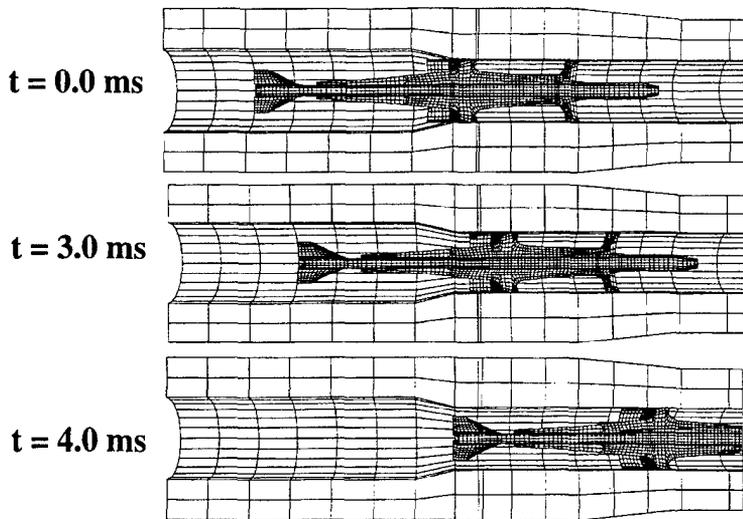


Figure 4: Kinetic energy projectile motion with nylon 6/6 obturator at $t = 0$ ms, $t = 3$ ms, and $t = 4$ ms (crosssectional view).

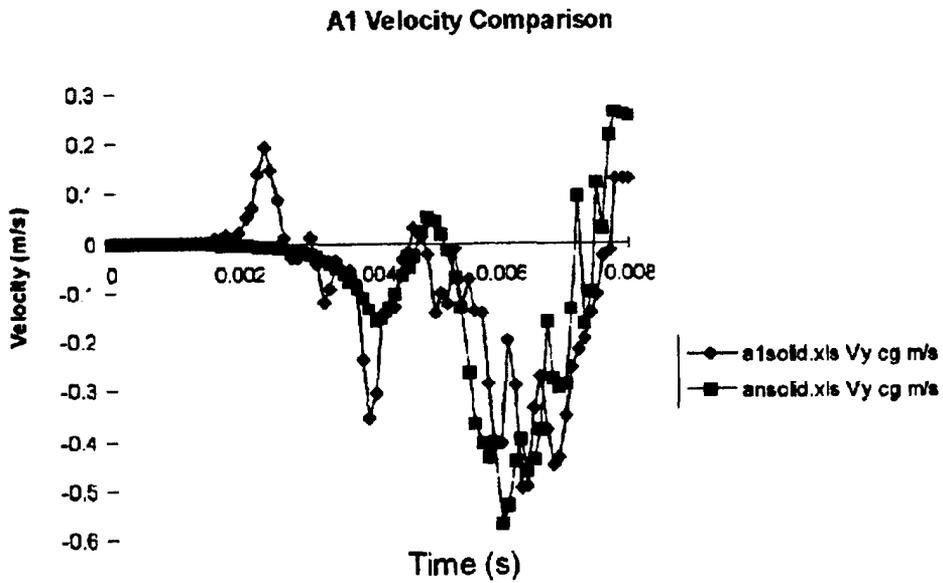


Figure 5: Vertical component of velocity history of projectile center of gravity, (cg); elastic (diamonds) vs nylon 6/6 (squares) obturator.

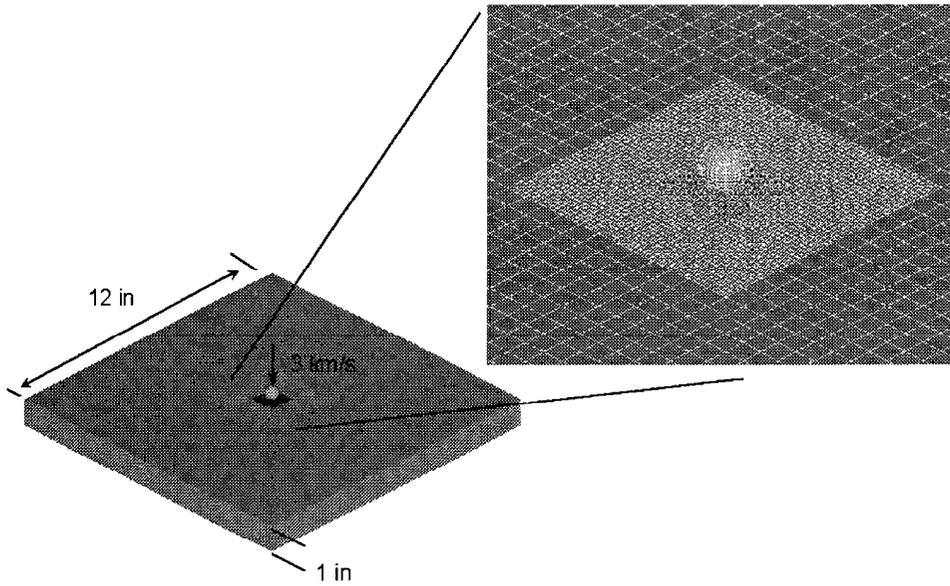


Figure 6: Steel ball penetration model in nylon 6/6 plate.

$$\bar{\epsilon}^p = 0.50$$

$$\bar{\epsilon}^p = 3.00$$

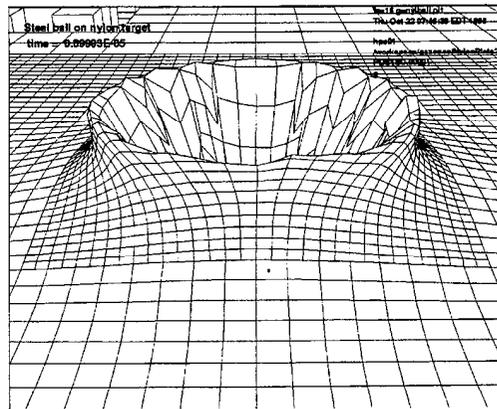
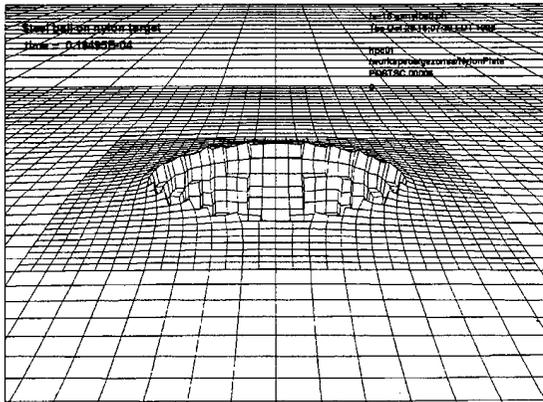


Figure 7: Influence of erosion strain $\bar{\epsilon}^p$ level on crater formation in a nylon 6/6 plate.

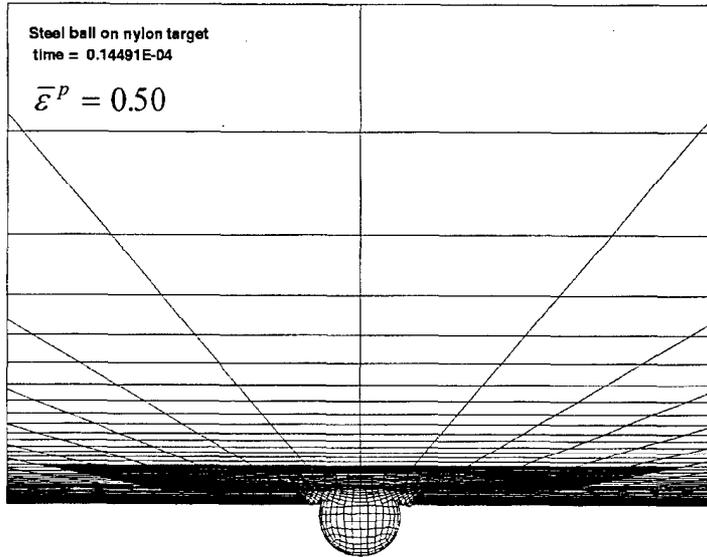


Figure 8: Steel ball exiting the bottom surface of a nylon 6/6 plate illustrating crater exit topology.

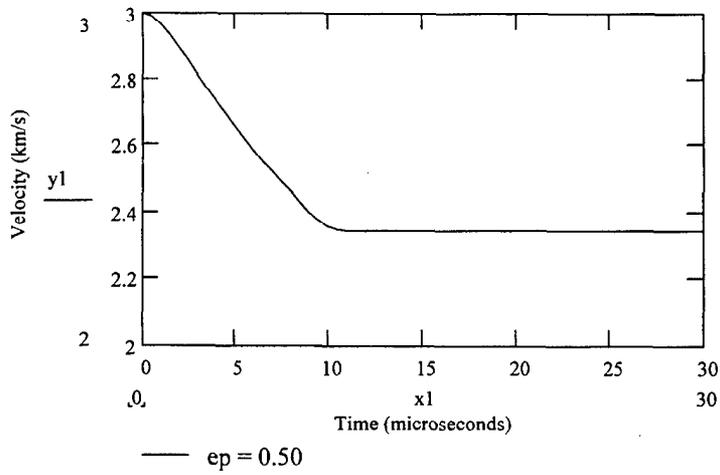


Figure 9: Velocity history of steel ball *cg* showing deceleration during penetration into a nylon 6/6 plate.

4 Summary

A rate and temperature dependent finite strain plasticity model for nylon 6/6 (Dupont Zytel® 101) that was originally developed for finite inelastic deformation in metals [2],[4], has been implemented in the explicit, Lagrangian finite element code DYNA3D [16]. The model has been previously implemented into the implicit code NIKE2D [10] for comparing the engagement process of nylon obturators relative to that of copper obturators [14]. Renewed interest in the mechanical behavior of nylon obturators [11] prompted this study, because of the materials' strongly hygroscopic behavior. Results were presented that demonstrated the validity of the nylon 6/6 model implementation with the closed-form results in [4], as well as implementation of a new Green-Naghdi stress rate in DYNA3D and its validation with the closed-form results in [9]. The importance of quantifying Lagrangian erosion strains were illustrated for problems involving friction, wear and penetration. The determination of proper user-defined erosion strains requires model calibration with experiments so that penetration metrics such as exit velocity and crater topology can be accurately modeled; such experiments are currently underway for nylon 6/6.

Another very important area of investigation would be the development of a microphenomenologically based internal state variable theory for the large-deformation of linear semicrystalline polymers much along the same lines as that presented by Bammann and Aifantis [3] for metals. Boyce [6] and Arruda et al [1] have made progress in this area and discuss microphysically based models for the large-deformation behavior of glassy polymers, such as PMMA. The theory is based upon both a scalar and second-order internal state variable formulation to describe polymer chain segment rotation and chain alignment, respectively, as microphysical mechanisms that contribute to plastic flow resistance in the polymer. Because the currently implemented model structure for nylon 6/6 is macrophysically based, this may pose severe limitations in the ability to extrapolate behavior in a predictive sense.

5 Acknowledgments

The author wishes to acknowledge Dr.'s Ken J. Perano and Paul E. Nielan (Sandia National Laboratories, Livermore, CA) for providing the nylon 6/6 material constants to the ARL, as well as Dr. Mike L. Chiesa (Sandia National Laboratories, Livermore, CA) for providing some useful coding suggestions relevant to the DYNA3D implementation. Dr. James F. Newill (US Army Research Laboratory, APG, MD) provided the mesh for the KE projectile simulations/analysis seen in Figures (4) and (5).

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REPORT DOCUMENTATION PAGE

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Public reporting burden for this collection of information is estimated to average 1 hour per response, including the time for reviewing instructions, searching existing data sources, gathering and maintaining the data needed, and completing and reviewing the collection of information. Send comments regarding this burden estimate or any other aspect of this collection of information, including suggestions for reducing this burden, to Washington Headquarters Services, Directorate for Information Operations and Reports, 1215 Jefferson Davis Highway, Suite 1204, Arlington, VA 22202-4302, and to the Office of Management and Budget, Paperwork Reduction Project (0704-0188), Washington, DC 20503.

1. AGENCY USE ONLY (Leave blank)		2. REPORT DATE September 2000	3. REPORT TYPE AND DATES COVERED Reprint, Oct 1999 to Jul 2000	
4. TITLE AND SUBTITLE Implementation of a Finite Strain Plasticity Model for Nylon 6/6 into DYNA3D			5. FUNDING NUMBERS DA PR: AH80 PE: 622618	
6. AUTHOR(S) George A. Gazonas				
7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES) U.S. Army Research Laboratory Attn: AMSRL-WM-MB email: gazonas@arl.army.mil Aberdeen Proving Ground, MD 21005-5069			8. PERFORMING ORGANIZATION REPORT NUMBER ARL-RP-1	
9. SPONSORING/MONITORING AGENCY NAME(S) AND ADDRESS(ES) U.S. Army Research Laboratory Aberdeen Proving Ground, MD 21005-5069			10. SPONSORING/MONITORING AGENCY REPORT NUMBER	
11. SUPPLEMENTARY NOTES ARL PR: AMS code: 622618H8011		A reprint from <i>Structures Under Shock and Impact VI</i> , N. Jones and C. A. Brebbia, editors, WIT Press (2000), pp 523-535.		
12a. DISTRIBUTION/AVAILABILITY STATEMENT Approved for public release; distribution unlimited.			12b. DISTRIBUTION CODE	
13. ABSTRACT (Maximum 200 words) A fully three-dimensional rate and temperature dependent finite strain plasticity model for semicrystalline nylon 6/6, originally developed for finite inelastic deformation in metals, has been implemented in the explicit, Lagrangian finite element code DYNA3D. The model has been previously implemented into the implicit code NIKE2D for comparing the engagement process of nylon obturators relative to that of copper obturators. The DYNA3D material driver was used to verify model implementation for constant strain rate input histories by comparing true stress versus true strain results with predictions from Bammann's closed-form model and NIKE2D implicit finite element results. The material model is being used for modeling the interaction and erosion of the nylon 6/6 obturator band against a smoothbore gun tube during launch in 120-mm tank rounds. The adaptive slideline feature also permits modeling complex impact and penetration phenomena by calibrating the Lagrangian element erosion strains against carefully controlled penetration experiments into nylon 6/6 plates.				
14. SUBJECT TERMS nylon 6/6, computational finite elements, finite deformation viscoplasticity, polymers, DYNA3D, impact, penetration, Green-Naghdi stress rate, Jaumann stress rate			15. NUMBER OF PAGES 32	
			16. PRICE CODE	
17. SECURITY CLASSIFICATION OF REPORT Unclassified	18. SECURITY CLASSIFICATION OF THIS PAGE Unclassified	19. SECURITY CLASSIFICATION OF ABSTRACT Unclassified	20. LIMITATION OF ABSTRACT UL	